

# Continuum Quantum Monte Carlo Methods

## Introduction, Recent Developments, Successes, Failures, and Prospects

W.M.C. Foulkes

Department of Physics  
Imperial College London

From Basic Concepts to Real Materials, KITP, 2009

# Outline

- 1 Continuum QMC Methods
- 2 Example Applications
- 3 The Good, The Bad and The Ugly
- 4 Recent Advances and Prospects

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# Bias

- I am an electronic structure/materials theorist.
  - ▶ Want real numbers for real materials.
  - ▶ More interested in solids than molecules.
  - ▶ More interested in ground states than excitations.
  - ▶ DFT is wonderful but not enough.
  - ▶ Lattice models are not enough.

# Bias

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  - ▶ Want real numbers for real materials.
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  - ▶ More interested in ground states than excitations.
  - ▶ DFT is wonderful but not enough.
  - ▶ Lattice models are not enough.
- Methods that interest me . . .
  - ▶ Get the electronic structure right first.
  - ▶ Describe correlation better than DFT.

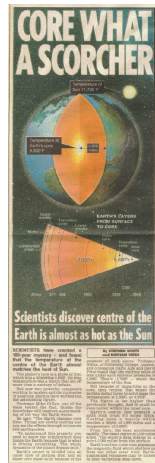
# The Earth's Core

- What is it?
- How hot is it?

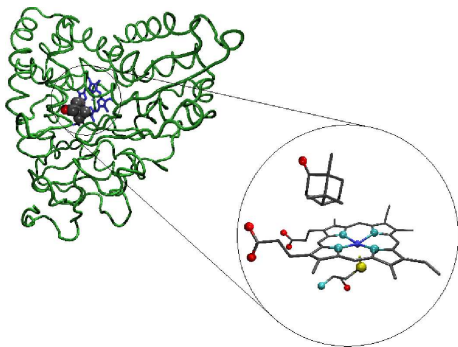
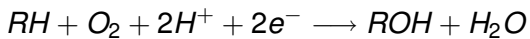
About 5500K at inner/outer core boundary

- How viscous it is?

Estimates span 12 orders of magnitude



# Cytochrome P450 Family of Enzymes



# Continuum QMC

Continuum QMC methods are

- Primarily for ground states/total energies.
- Are not perturbative (variational instead).
- Are systematically improvable.
- Useful in large systems.



# Continuum QMC

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- Are systematically improvable.
- Useful in large systems.

Tackle the many-electron problem head-on!

[The many-electron problem is (NP-)hard!]

# The Many-Electron Schrödinger Equation

## Textbook Notation

$$\left( \sum_{i=1}^N \left\{ -\frac{\hbar^2}{2m} \nabla_i^2 + v_{\text{ext}}(\mathbf{r}_i) \right\} + \sum_{i>j} \frac{e^2}{4\pi\epsilon_0 |\mathbf{r}_i - \mathbf{r}_j|} \right) \Psi(\mathbf{r}_1, \dots, \mathbf{r}_N) = E\Psi(\mathbf{r}_1, \dots, \mathbf{r}_N)$$

# The Many-Electron Schrödinger Equation

## Textbook Notation

$$\left( \sum_{i=1}^N \left\{ -\frac{\hbar^2}{2m} \nabla_i^2 + v_{\text{ext}}(\mathbf{r}_i) \right\} + \sum_{i>j} \frac{e^2}{4\pi\epsilon_0 |\mathbf{r}_i - \mathbf{r}_j|} \right) \Psi(\mathbf{r}_1, \dots, \mathbf{r}_N) = E\Psi(\mathbf{r}_1, \dots, \mathbf{r}_N)$$

## Simpler Notation

$$\left\{ -\frac{1}{2} \nabla_{\mathbf{R}}^2 + V(\mathbf{R}) \right\} \Psi(\mathbf{R}) = E\Psi(\mathbf{R})$$

$$\mathbf{R} = (\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N)$$
$$\nabla_{\mathbf{R}} = (\nabla_{\mathbf{r}_1}, \nabla_{\mathbf{r}_2}, \dots, \nabla_{\mathbf{r}_N})$$

Atomic units:  $e = \hbar = m = 4\pi\epsilon_0 = 1$   
Unit of length =  $a_0 = 0.529177 \text{ \AA}$   
Unit of energy = Hartree = 27.211 eV

# Energy Scales

Total electronic	$> 10^2 \text{ eV}$	$(> 10^3 \text{ eV})$
Chemical bond	few eV	
Chemical reaction	$< 10^{-1} \text{ eV}$	
Room temperature	$2.5 \times 10^{-2} \text{ eV}$	
High $T_c$ superconductivity	$< 10^{-2} \text{ eV}$	

## C atom

$$E_{\text{total}} = -1075 \text{ eV}$$

$$E_{\text{HF}} = 99.6\% E_{\text{total}} = -1070.7 \text{ eV}$$

$$E_{\text{corr}} = 0.4\% E_{\text{total}} = -4.3 \text{ eV}$$

## C pseudo-atom

$$E_{\text{total}} = -150 \text{ eV}$$

$$E_{\text{HF}} = 98.2\% E_{\text{total}} = -147.3 \text{ eV}$$

$$E_{\text{corr}} = 1.8\% E_{\text{total}} = -2.7 \text{ eV}$$

# Continuum QMC

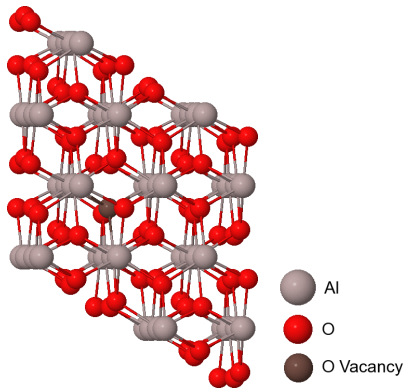
The

- variational
- diffusion
- reptation, path-integral, phaseless auxiliary-field, . . .

QMC methods take the HF approximation as a starting point and add a reasonable description of correlation.

- Not accurate enough to describe subtle correlation physics ( $< 10^{-2}$  eV).
- Can attain chemical accuracy ( $1 \text{ kcal mole}^{-1} \sim 0.04 \text{ eV}$ ) in molecules.
- Can answer most questions about chemical bonding.
- About 10 times (?) better than DFT (but  $\gg$  10 times the effort).

- $N^3$  (or better) scaling with system size.
- Can study systems of 1000+ electrons.



576 valence electrons (without vacancy)



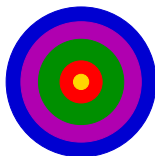
# The VMC Algorithm

- Guess  $\Psi_T(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_{10^{23}}) = \Psi_T(\mathbf{R})$ .
- Evaluate

$$\begin{aligned} E_T &= \int \Psi_T^*(\mathbf{R}) \hat{H} \Psi_T(\mathbf{R}) d\mathbf{R} \\ &= \int \left( \frac{\hat{H} \Psi_T(\mathbf{R})}{\Psi_T(\mathbf{R})} \right) |\Psi_T(\mathbf{R})|^2 d\mathbf{R} \end{aligned}$$

using Monte Carlo integration.

- Probability density =  $|\Psi(\mathbf{R})|^2$ ; score =  $E_L(\mathbf{R}) = \frac{\hat{H}\Psi(\mathbf{R})}{\Psi(\mathbf{R})}$ .



# Slater-Jastrow Trial Functions

**Strategy:** first get mean-field right, then add a reasonable description of correlation:

$$\Psi(\mathbf{R}) = \exp \left[ - \sum_{i>j} u(\mathbf{r}_i, \mathbf{r}_j) + \sum_i \chi(\mathbf{r}_i) \right] \sum_{\alpha} c_{\alpha} D_{\alpha}(\mathbf{R}),$$

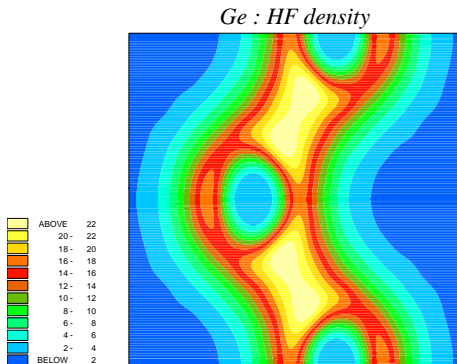
where  $\alpha = (i_1, i_2, \dots, i_N)$  and

$$D_{\alpha}(\mathbf{R}) = \begin{vmatrix} \psi_{i_1}(\mathbf{r}_1) & \psi_{i_1}(\mathbf{r}_2) & \dots & \dots & \psi_{i_1}(\mathbf{r}_N) \\ \psi_{i_2}(\mathbf{r}_1) & \psi_{i_2}(\mathbf{r}_2) & \dots & \dots & \psi_{i_2}(\mathbf{r}_N) \\ \dots & \dots & \dots & \dots & \dots \\ \dots & \dots & \dots & \dots & \dots \\ \psi_{i_N}(\mathbf{r}_1) & \psi_{i_N}(\mathbf{r}_2) & \dots & \dots & \psi_{i_N}(\mathbf{r}_N) \end{vmatrix}.$$

# Slater-Jastrow Trial Functions

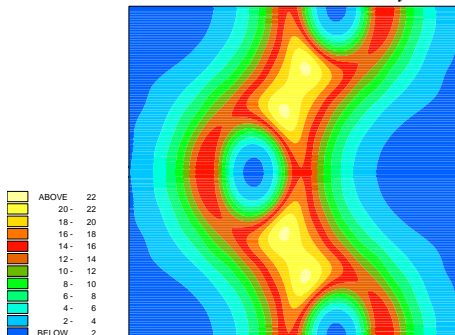
- Orbitals  $\psi_i(\mathbf{r})$  from DFT or Hartree-Fock.
- $u(\mathbf{r}_i, \mathbf{r}_j)$  increases when  $\mathbf{r}_i \rightarrow \mathbf{r}_j$ . Helps keep electrons apart (correlation).  $\chi(\mathbf{r}_i)$  counteracts smearing effect of  $u$ .
- $u$  and  $\chi$  given flexible variational representations.
- Expansion coefficients  $c_\alpha$  treated as variational parameters.
- Orbitals in determinants may be optimized too.

# HF Density

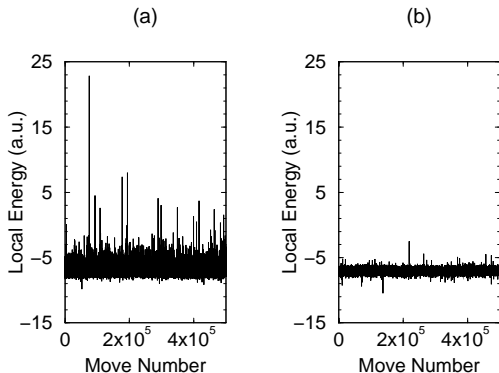


# HFJCHI Density

*Ge : HFJ + CHI density*



# Effect of Jastrow Factor in Silane Molecule



# The Cusp Conditions

- As any two electrons approach one another, both terms in

$$E_L(\mathbf{R}) = \frac{-\frac{1}{2}\nabla_{\mathbf{R}}^2\Psi(\mathbf{R})}{\Psi(\mathbf{R})} + V(\mathbf{R})$$

diverge.

- ▶ In exact ground state, divergences cancel exactly.
- ▶ For Slater determinant, KE term does not diverge.
- ▶ For SJ trial function, divergences cancel if  $u(r_{ij})$  obeys the *cusp conditions*:

$$\left. \frac{\partial u(r_{ij})}{\partial r_{ij}} \right|_{r_{ij}=0} = \begin{cases} -\frac{1}{2} & \text{parallel spins} \\ -\frac{1}{4} & \text{antiparallel spins} \end{cases}$$

# Multiple Determinants

In molecules with strong static correlation, it is common to use trial functions including hundreds (but not millions) of Slater determinants.

In solids, this is less common:

- Suppose you have  $N$  electrons and use  $2N$  basis functions in your mean-field calculation.
- Could construct  ${}^{2N}C_N \approx e^{(2 \ln 2)N}$  determinants!
- Which to choose?

Fortunately, one-determinant SJ trial functions work surprisingly well in weakly correlated solids. Typically account for 85%+ of correlation energy.



# Advantages and Disadvantages

- Advantages:

- ▶ Surprising that such a simple approach returns 85% of  $E_c$ .
- ▶ Relatively quick (only 10 times DFT!)

- Disadvantages:

- ▶ 85% of  $E_c$  is not enough.
- ▶ Hard to ensure error cancellation.
- ▶ Wave function optimization is a pain.
- ▶ Trial functions for  $\text{La}_2\text{CuO}_4$ ?
- ▶ Garbage in — garbage out!

# Diffusion QMC

- The DMC method overcomes many of the problems of VMC.
- Uses a stochastic projection technique to improve an initial trial function (often taken from a prior VMC calculation).
- Typically returns 95%+ of correlation energy.
- DMC and affiliates are much the most accurate total energy methods available for large systems.

# The Imaginary-Time Schrödinger Equation

$$\begin{aligned}\frac{\partial \Psi(t)}{\partial t} &= -\hat{H}\Psi(t) \\ \frac{\partial \Psi(\mathbf{R}, t)}{\partial t} &= \left[ \frac{1}{2} \nabla_{\mathbf{R}}^2 - V(\mathbf{R}) \right] \Psi(\mathbf{R}, t)\end{aligned}$$

Cross between a  $3N$ -dimensional diffusion equation

$$\frac{\partial n(\mathbf{R}, t)}{\partial t} = D \nabla_{\mathbf{R}}^2 n(\mathbf{R}, t)$$

and an exponential decay equation

$$\frac{\partial N}{\partial t} = -\kappa N$$

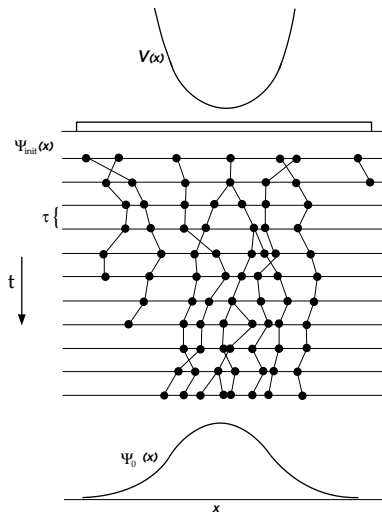
with a position-dependent decay rate.

# Simple DMC Algorithm

## Simple DMC Algorithm

- Scatter walkers  $\mathbf{R}$  through configuration space. Each walker has initial “weight”  $w = 1$ .
- In time  $\Delta t$ , walker at  $\mathbf{R}$ :
  - ▶ moves to  $\mathbf{R}' = \mathbf{R} + \xi\sqrt{2D\Delta t} = \mathbf{R} + \xi\sqrt{\Delta t}$ ,
  - ▶ changes weight from  $w$  to  $w' = we^{-V(\mathbf{R})\Delta t}$ .
- To prevent walkers accumulating wild weights, allow low-weight walkers to die out and high-weight walkers to split up, keeping expected weight at every point unchanged.

# DMC Evolution



# The Sign Problem

Since  $\psi(\mathbf{R})$  is antisymmetric, it cannot be a probability density!

DMC algorithm works, but yields the nodeless many-boson ground state of  $\hat{H}$ .

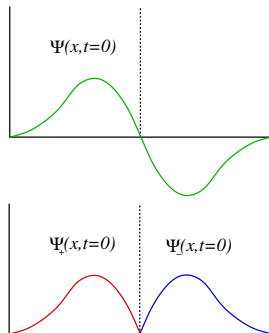
Can we circumvent this problem?

# First Excited State of a Particle in a Box

- At  $t = 0$ , write  $\Psi = \Psi_+ - \Psi_-$ , where

$$\Psi_+ = \frac{1}{2}(|\Psi| + \Psi), \quad \Psi_- = \frac{1}{2}(|\Psi| - \Psi).$$

- Evolve positive distributions  $\Psi_+(t)$  and  $\Psi_-(t)$  separately.
- Antisymmetric ground state  
 $= \lim_{t \rightarrow \infty} [\Psi_+(t) - \Psi_-(t)].$

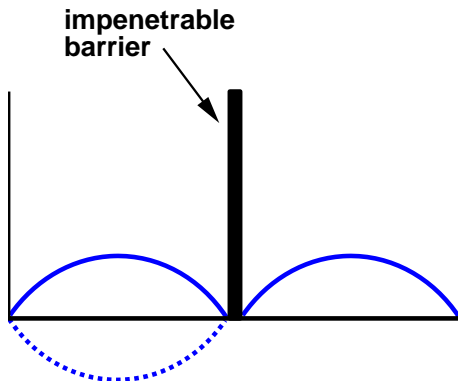


# Failure!

- Unfortunately, both  $\Psi_+$  and  $\Psi_-$  tend to the symmetric ground state as  $t \rightarrow \infty$ .
- The antisymmetric components decrease like  $e^{-(E_1-E_0)t}$  in comparison.
- The subtraction required to extract the antisymmetric component becomes more and more difficult as  $t$  increases and the result is soon swamped by the statistical noise.



# The Fixed-Node Approximation for the First Excited State



# The Fixed-Node Approximation for the Many-Electron Ground State

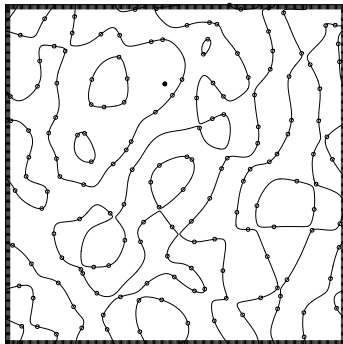
## The Fixed-Node DMC Algorithm

- Take an antisymmetric trial wavefunction (from VMC).
- Scatter walkers as normal and start DMC dynamics.
- Put impenetrable barriers on the nodal surface.
- Delete any walker attempting a trial move for which

$$\Psi_T(\mathbf{R}_{\text{new}})/\Psi_T(\mathbf{R}_{\text{old}}) < 0 .$$

- The fixed-node approximation is variational and normally very accurate (> 95% of correlation energy).
- We know of nothing better to do in large systems.

# Nodal Surface



A 2D slice through the 321-dimensional nodal surface of a gas of 161 spin-up electrons. The positions of 160 electrons are fixed and the nodes plotted as a function of the position of the 161<sup>st</sup>.

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# The G1 Set

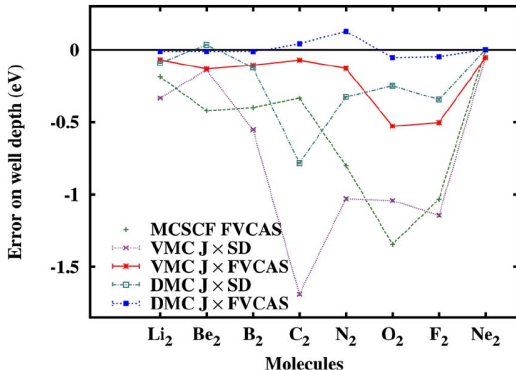
- Simplest plausible approach:
  - ▶ one determinant
  - ▶ no orbital optimization
- Mean absolute deviation: 2.9 kcal/mol (0.13 eV). (Comparable to B3LYP and CCSD(T) with similar basis.)
- Maximum deviation ( $\text{SO}_2$ ): 14 kcal/mol (0.61 eV).

Not great, but remember that DMC *scales*

[J.G. Grossman, J. Chem. Phys. **117**, 1434 (2002)]

# Can We Do Better?

- Well depths for first-row homonuclear diatomic molecules.
- Fully optimized  $J \times FVCAS$  trial function with up to  $\sim 150$  CSFs.
- Typical accuracy 1 kcal/mol = 0.04 eV; largest error ( $N_2$ ) 0.1 eV.
- $Be_2$  and  $Ne_2$  OK.



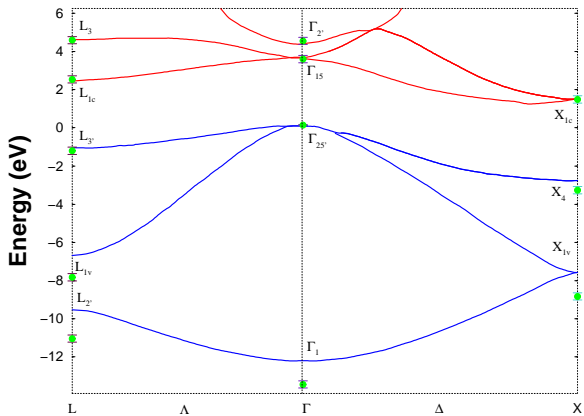
[J. Toulouse and C.J. Umrigar, J. Chem. Phys. **128**, 174101 (2008).]

# Cohesive Energies

Method	Si	Ge	C	BN	Al
Experiment	4.62	3.85	7.37	12.9	3.43
DFT	5.28	4.59	8.61	15.07	4.21
VMC	4.48	3.80	7.36	12.85	3.23
DMC	4.63	3.85	7.46		3.47

in eV per atom  
statistical errors typically  $\pm 0.04$  eV/atom

# Band Structure of Si



[A.J. Williamson, R.Q. Hood, R.J. Needs, and G. Rajagopal, Phys. Rev. B **57**, 12140 (1998).]



# Finding out about Exchange and Correlation

## The Adiabatic Connection Formula

$$E_{\text{xc}}[n] = \frac{1}{2} \iint \frac{n(r)\bar{n}_{\text{xc}}(r, r')}{|r - r'|} dr dr'$$

where

$$\bar{n}_{\text{xc}}(r, r') = \int_{\lambda=0}^1 n_{\text{xc}}^{\lambda}(r, r') d\lambda$$

$$\hat{H}^{\lambda}\psi^{\lambda} = \left( \hat{T} + \lambda \hat{V}_{\text{ee}} + \hat{V}^{\lambda} \right) \psi^{\lambda} = E^{\lambda}\psi^{\lambda}$$

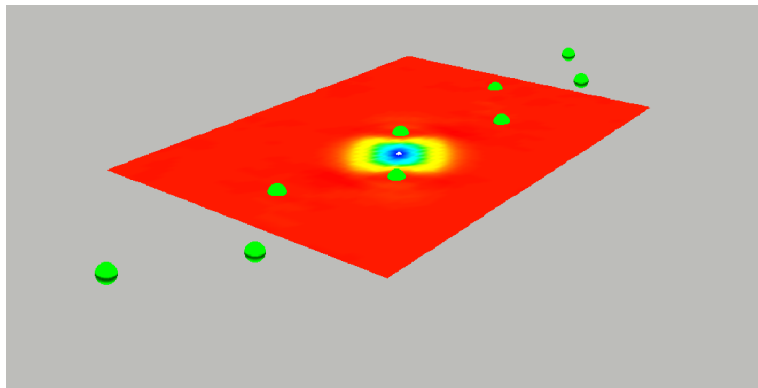
$$n^{\lambda}(r) = n(r)$$

# VMC Realisation of the Adiabatic Connection

$$E_{\text{xc}} = \frac{1}{2} \iint \frac{n(r)\bar{n}_{\text{xc}}(r, r')}{|r - r'|} dr' dr = \int e_{\text{xc}}(r, [n]) dr$$

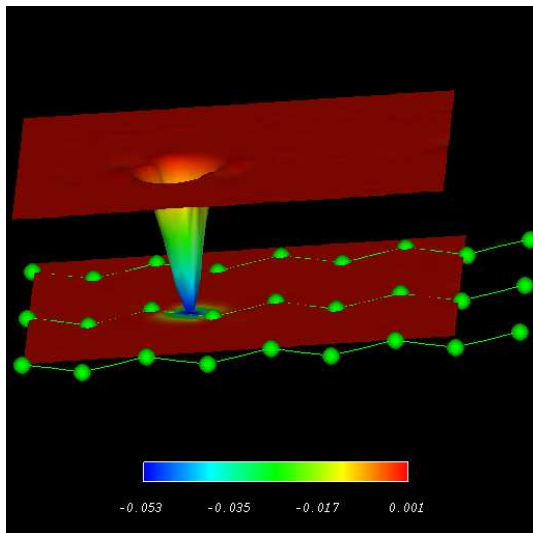
- Slater-Jastrow ansatz for many-body wavefunction.
- Integral over  $\lambda$  discretised.
- Variational parameters in  $\psi^\lambda$  and Fourier components of  $V^\lambda$  determined at every  $\lambda$ .
- Statistical errors small.
- Tried very hard to account for all the systematic errors (particularly finite-size errors).

# The Exchange-Correlation Hole in Si

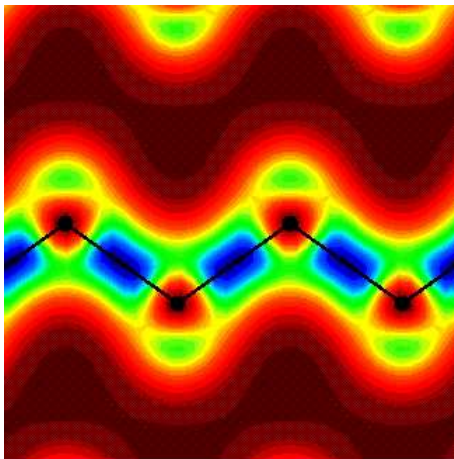


[R.Q. Hood, M.-Y. Chou, A.J. Williamson, G. Rajagopal, and R.J. Needs,  
Phys. Rev. B 57, 8972 (1998).]

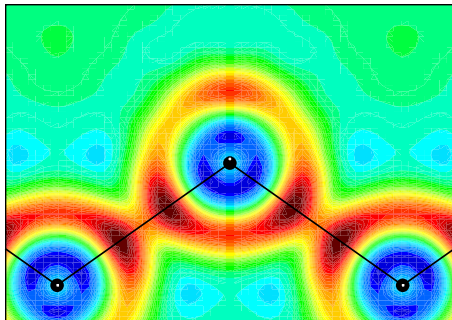
# The Exchange-Correlation Hole in Si



# Exchange-Correlation Energy Density in Silicon



# Error in LDA Exchange-Correlation Energy Density in Silicon



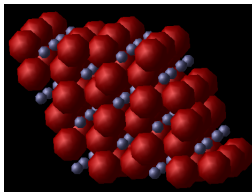
# Point Defects in Alumina

Same mineral, different defects: Corundum, Ruby and Sapphire



Cr doped

Ti/Fe doped



N.D.M. Hine, K. Frensch, M.W. Finnis, W.M.C. Foulkes, and A.H. Heuer,  
Phys. Rev. B **79**, 024112 (2009) and tbp.

# Formation Energies

- For defects in alumina of charge  $q = -\Delta n_e$ ,

$$E_f = G_{def}^q - G_{perf} - \Delta n_{Al} \mu_{Al} - \Delta n_O \mu_O + q \mu_e.$$

- $G_{def}^q$  and  $G_{perf}$  approximated by total energies plus vibrational contributions.
- $\mu_O, \mu_{Al}$  fixed by annealing conditions.



# Why DMC?

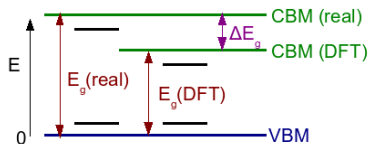
**Table:** Formation (all in eV).  $\Delta H_0^{\text{Al}}$  and  $\Delta H_0^{\text{O}}$  are the formation enthalpies per atom of Al and O atoms in the gas phase.  $\Delta H_0^{\text{AlO}}$  is the formation energy of an AlO molecule.

Method	$\Delta H_0^{\text{Al}}$	$\Delta H_0^{\text{O}}$	$\Delta H_0^{\text{AlO}}$
LDA-USP	4.05	3.62	0.91
LDA-DF	4.10	3.67	1.13
GGA-USP	3.41	2.82	0.74
DMC	3.47(1)	2.54(1)	0.68(1)
Experiment	3.42	2.58	0.69

# Why DMC?

In case this is not yet convincing . . .

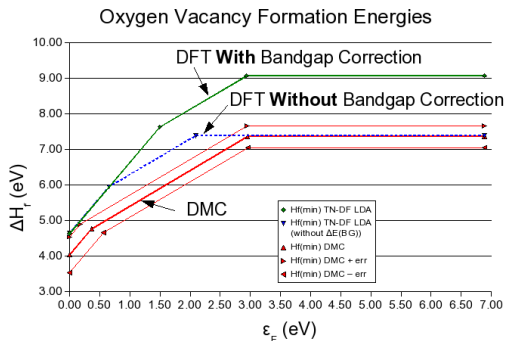
Occupied defect states deriving from conduction band states are too low in energy because of DFT gap underestimation.



Correction  $m \times \Delta E_g$  is normally applied to DFT formation energies.

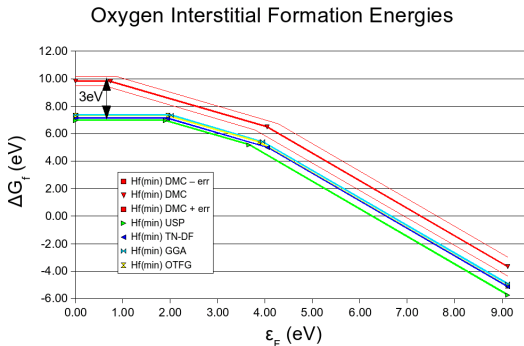
# DMC Formation Energies

If no bandgap correction is applied, DMC results agree well with DFT except for correcting self-interaction error of localised states.



DFT (including bandgap correction) appears to be significantly overbinding. Real cost to break bonds is lower.

# DMC Formation Energies



Interstitial is consistently easier to form in DFT, also suggesting DFT overbinds.

## Other Applications

- Surface energies and reactions at surfaces.
- Clusters.
- Static linear response.
- Fermi liquid parameters.
- Hydrogen bonding and van der Waals bonding.
- Transition metal oxides.
- Metal-insulator transitions.
- Phase stability.
- ...

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# The Good

- DMC works well — stable and nearly automatic once it is going.
- DMC algorithm is “trivially parallel”.
- More accurate than DFT. Normally much more accurate.
- The only (?) way to benchmark DFT in large systems.
- System-size scaling no worse than DFT.
- Provides direct information about correlation.
- Excellent total energies; often lower than full CI.

# The Bad

- DFT is  $10^3$ – $10^4$  (?) times faster. DMC is currently too slow for serious quantum molecular dynamics.
- Excellent total energies do not imply similarly excellent energy differences: fixed-node errors do not cancel as well as you might hope.
- Limited ideas about how to improve nodes in large systems.
- Very limited information about excited states.
- Force calculations are hard and slow.



# The Ugly

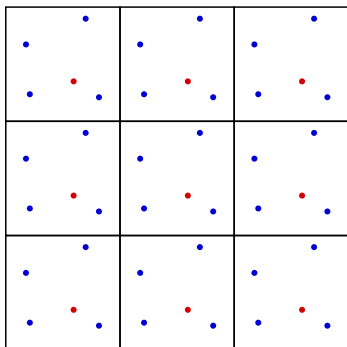
- Practicality relies on pseudopotentials:
  - ▶ Not a many-electron concept.
  - ▶ Treatment of non-locality requires additional approximation.
- Trial function construction and optimization is a pain.
- Finite-size errors are a pain.

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## Finite-Size Errors

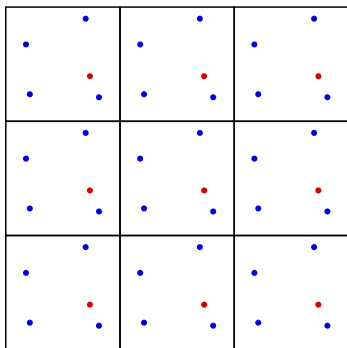
... are the largest errors in most QMC calculations of solids.



Thanks to the work of Ceperley, Chiesa, Drummond, Foulkes, Fraser, Holzmann, Kwee, Needs, Williamson, Zhang, . . . , these errors are now under much better control.

## Finite-Size Errors

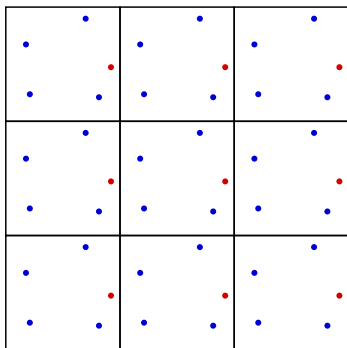
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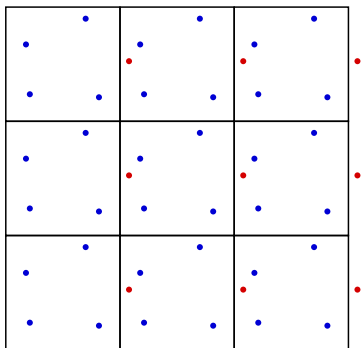
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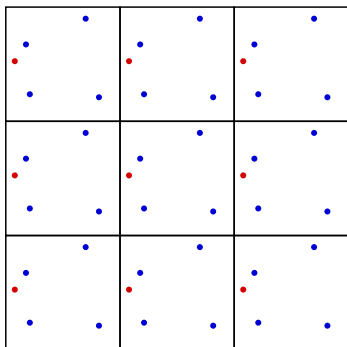
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Thanks to the work of Ceperley, Chiesa, Drummond, Foulkes, Fraser, Holzmann, Kwee, Needs, Williamson, Zhang, . . . , these errors are now under much better control.

## Finite-Size Errors

... are the largest errors in most QMC calculations of solids.



Thanks to the work of Ceperley, Chiesa, Drummond, Foulkes, Fraser, Holzmann, Kwee, Needs, Williamson, Zhang, . . . , these errors are now under much better control.

# Trial Wavefunction Optimization

- Filippi, Nightingale, Sorella, Toulouse, Trail, Umrigar, . . . , have enormously improved wavefunction optimization techniques over the past few years.
  - ▶ Automatic optimization (almost) a reality.
  - ▶ No more need to sacrifice graduate students.

This is *much* more important than it sounds



# Better Trial Functions for Large Systems

Two main ideas are being investigated.

## Geminals and Pfaffians

Attacalite, Bajdich, Casula, Mitas, Schmidt, Sorella, Wagner.

Replace determinant by antisymmetrized product of pairing functions:

$$\hat{A}\phi_1(\mathbf{x}_1, \mathbf{x}_2)\phi_2(\mathbf{x}_3, \mathbf{x}_4) \dots \phi_{N/2}(\mathbf{x}_{N-1}, \mathbf{x}_N).$$

- If all  $\phi$  functions are identical singlets, get BCS wavefunction.
- If  $\phi(\mathbf{x}_1, \mathbf{x}_2) = \sum_{k=1}^N \phi_k(\mathbf{x}_1)\phi_k(\mathbf{x}_2)$ , the BCS wavefunction reduces to a Slater determinant.
- Can combine singlet pairing, triplet pairing and unpaired electrons.

## Backflow

Ceperley, Esler, Drummond, Holzmann, Kwon, Lin, López-Ríos, Ma, Martin, Needs, Pierleoni, Towler, Zong.

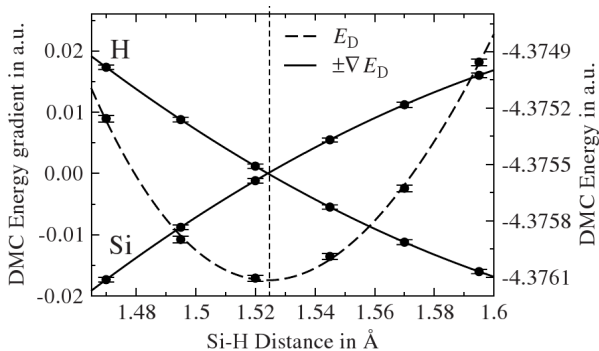
Electron positions in determinants are replaced by collective coordinates:

$$D(\mathbf{R}) \rightarrow D(\tilde{\mathbf{R}}),$$

where  $\tilde{\mathbf{r}}_i = \mathbf{r}_i + \boldsymbol{\xi}_i(\mathbf{R})$ .

# Forces

- Difficult because the nodal constraint (which is in effect a boundary condition on the wave function) depends on nuclear positions. Leads to hard-to-evaluate Pulay-like terms.
- Combining recent advances in low-variance estimators (Assaraf, Caffarel, Filippi, Umrigar, ...) with the Hellmann-Feynman theorem allowed Badinski, Needs and co-workers to evaluate forces accurately (although approximately). Effort remains large.



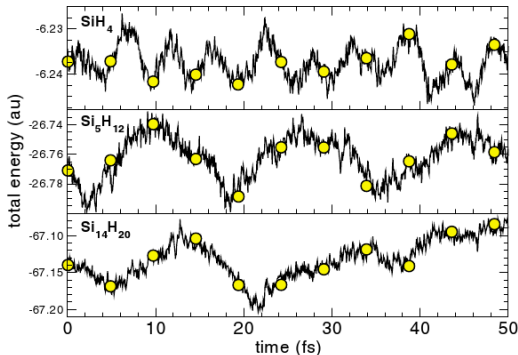
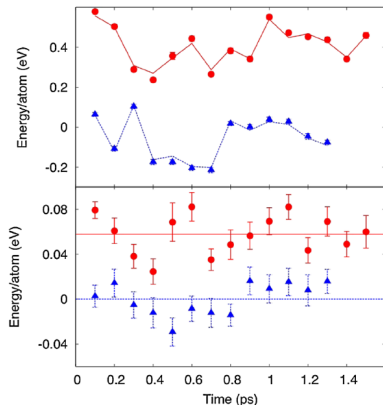


FIG. 1 (color online). DMC total ground state energies vs time for 5 fs time intervals of an AIMD simulation of SiH<sub>4</sub>, Si<sub>5</sub>H<sub>12</sub>, and Si<sub>14</sub>H<sub>20</sub> at 1000 K. The circles (lines) correspond to discretely sampled (continuous) DMC calculations. The error bars of the discretely sampled data are smaller than the symbol size.

[J.C. Grossman and L. Mitas, Phys. Rev. Lett. **94**, 056403 (2005).]

# The Melting Curve of Iron

$$\Delta F = \int_0^1 d\lambda \langle \Delta U \rangle_\lambda \approx \langle \Delta U \rangle_{\lambda=0} - \frac{1}{2k_B T} \left\langle (\Delta U - \langle \Delta U \rangle_{\lambda=0})^2 \right\rangle_{\lambda=0}$$



Solid is **blue**  
Liquid is **red**

$P = 330\text{GPa}$

$T_{\text{DFT}}^{\text{m}} = 6350 \pm 300\text{K}$

$\Delta T^{\text{m}} = 550 \pm 230\text{K}$

[E. Sola and D. Alfè, Phys. Rev. Lett. **103**, 078501 (2009).]

# New QMC Algorithms

## Phaseless Auxiliary-Field QMC

S. Zhang, in *Theoretical Methods for Strongly Correlated Electron Systems*, D. Senechal and A.-M. Tremblay, Eds. (Springer-Verlag, 2003).

## FCI-QMC

G.H. Booth, A.J.W. Thom, and A. Alavi, *J. Chem. Phys.* **131**, 054106 (2009).

# Challenges

- “Real” QMC MD.
- “Real” excitations.
- Stronger correlations.
- Systematic improvement of nodal surface.
- Larger systems and more real applicaitons (no more dimers!).

# Summary

- QMC is useful but field is small and relatively underdeveloped.
  - ▶ Still room for fundamental methods development.
  - ▶ Still room to make a big splash.
- QMC is perfect for next generation hardware. QMC MD is on the way.
- QMC is not the one true answer: you cannot calculate everything and you have to work hard to get results.
- After a slow patch, techniques have advanced substantially during the past few years, leaving me optimistic about the future.